

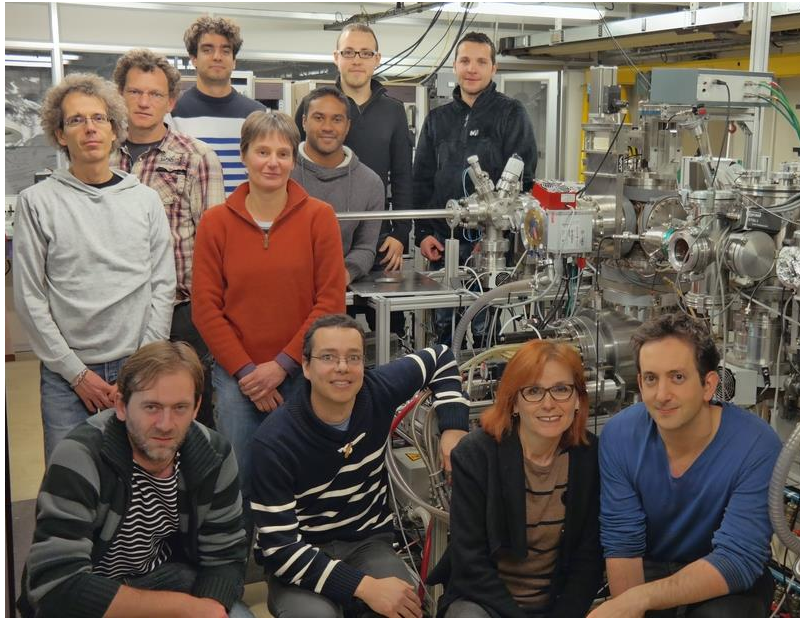
Magnetic cluster deposition: from individual nano-objects to complex systems

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Institut Lumière Matière, F-69622, LYON**



“Magnetic nanostructures” group



Small team (5 permanent researchers),
led by Véronique Dupuis

- ✓ Cluster deposition
- ✓ Magnetic nanoparticles
- ✓ SQUID magnetometry and modelling
- ✓ Microscopy (STM, TEM) and many other techniques (x-rays...)

Model systems

➡ Characterization of the intrinsic properties of nanomagnets

Nanomagnets as building blocks (bottom up approach)

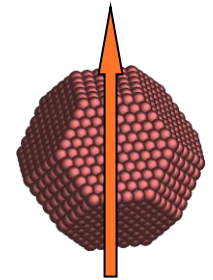
➡ More complex systems (clusters on surfaces, nano-composites)

Outline

- Motivation and experimental approach of cluster deposition
- Specific behavior of bimetallic nanoparticles
- Analysis of magnetic measurements: intrinsic properties with diluted nanomagnet assemblies and beyond...
- Interface effects with nanomagnets on cristaline surfaces: FePt/graphene/Ir(111) and FeRh/BaTiO₃
- Nano-composites: Co clusters in a FePt matrix

**Our interest: small nanoparticles,
between 1 nm and 10 nm diameter**

- ✓ Size reduction effects (importance of surface/interface)
- ✓ Miniaturization and new properties (multi-functional, exotic objects...)



Monodomain particle
(macrospin)

General question: link between structure (at the atomic and nanometer scale) and magnetism?

➡ How can the magnetic properties of nanostructures be controlled and tailored?

Fundamental research ...and interesting for **potential applications...**



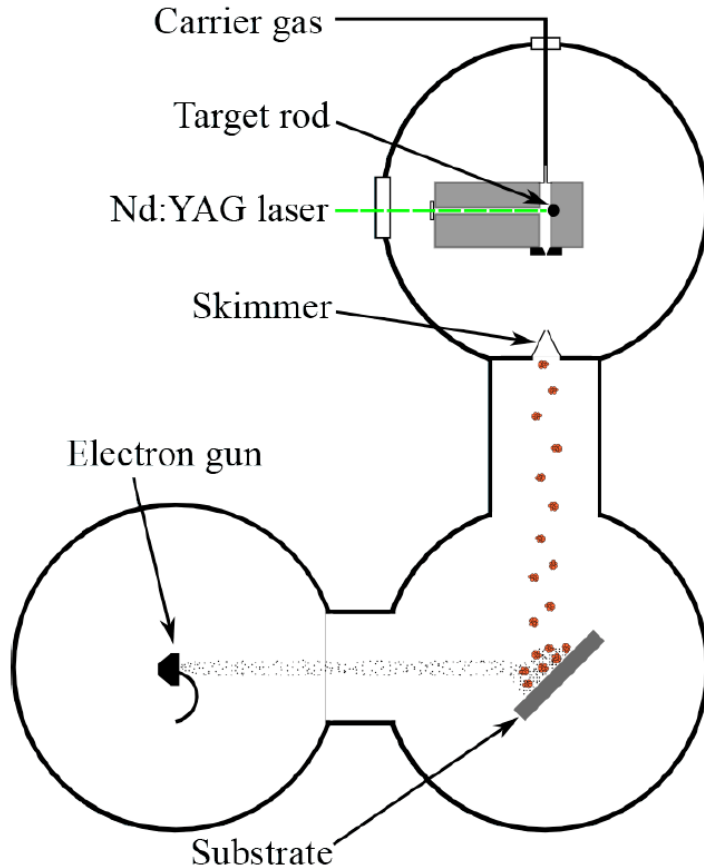
- Biological/medical applications
- Catalysis (transition metals and alloys)
- Magnetic storage applications
- Spintronic devices?

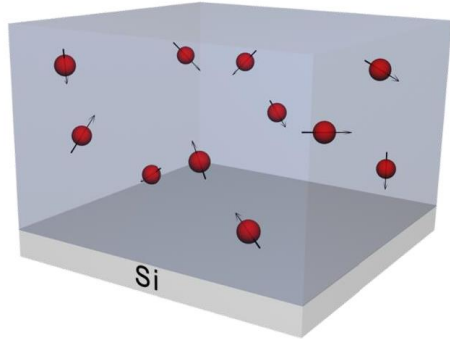
Synthesis challenge ➡ Well-defined nanoparticle samples

Deposition of preformed clusters (physical route)

Low energy cluster beam deposition, based on a laser vaporization source

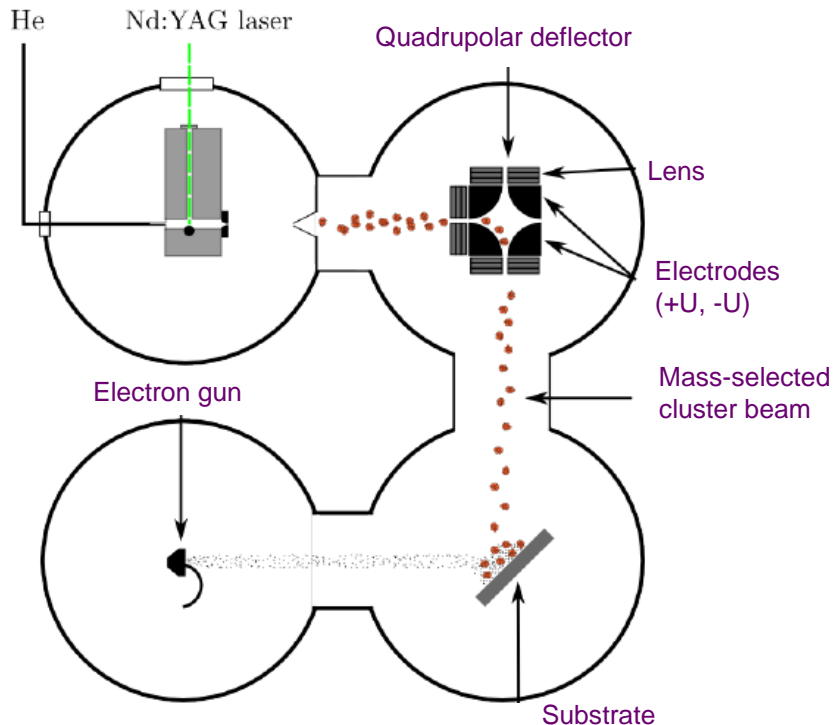
- ✓ Deposition under ultra-high vacuum
- ✓ Pure elements or alloys
 - ➡ Adjustable composition (target)
- ✓ Capping or co-deposition in a matrix
 - ➡ • Protect the particles
 - ➡ • Avoid coalescence
- ✓ Random deposition
 - ➡ Interparticle distances distribution directly fixed by the dilution





- ✓ Deposition on any substrate
- ✓ 2D, multilayers or 3D samples
(thin films of particles in a matrix)

This approach allows ex-situ characterization by many techniques (EXAFS, XRD, XMCD, TEM, SQUID...)



- ✓ Possibility of size selection
(quadrupolar electrostatic deflector)

All the particles have the same velocity
 → Selection of kinetic energy = mass selection

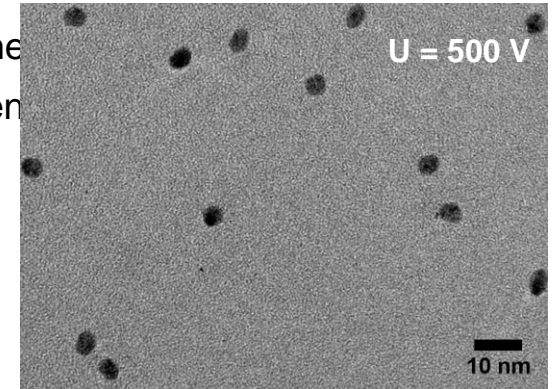
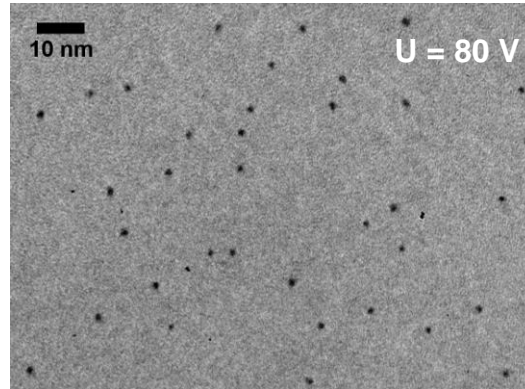
- Adjustable particle size, independently from the surface density.



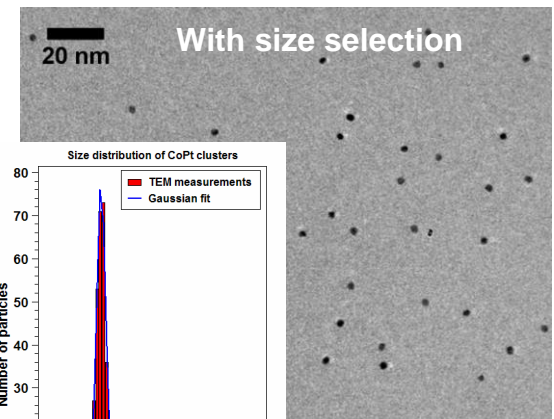
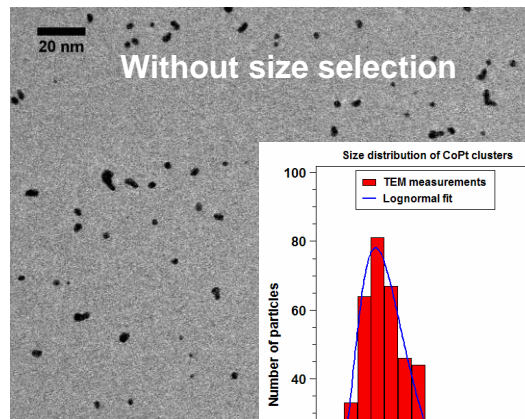
Diluted assemblies
(avoid interactions)

- ✓ Typical concentration for 3D samples ~1% in volume

- Relative diameter dispersion lower than 10 % with size selection.

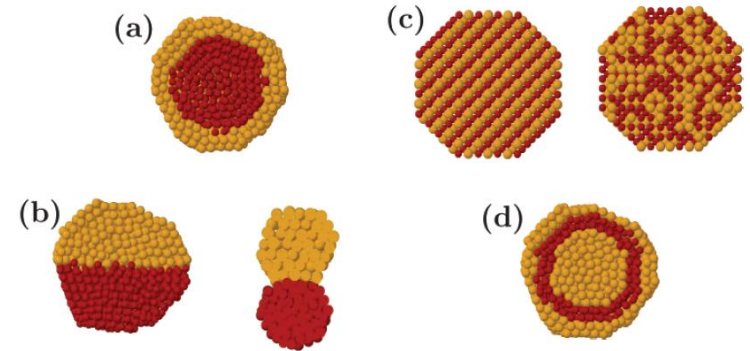


CoPt nanoparticles



$$\Delta D/D_m \sim 7-8 \%$$

- ➔ Two types of atoms:
additional degree of freedom
- ➔ Nanoalloys, bimetallic particles:
different types of structures

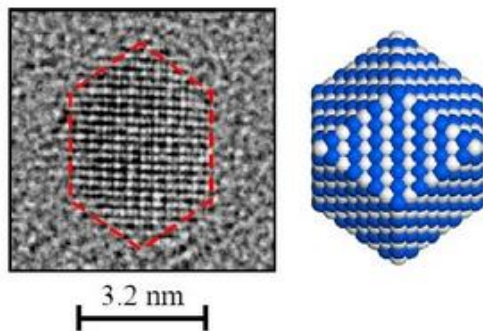


New properties, combination of properties, at the nanoscale

Many bi-metallic systems can be produced by LECBD

V. Dupuis *et al.*, *Phys. Chem. Chem. Phys.* **17**, 27996 (2015).

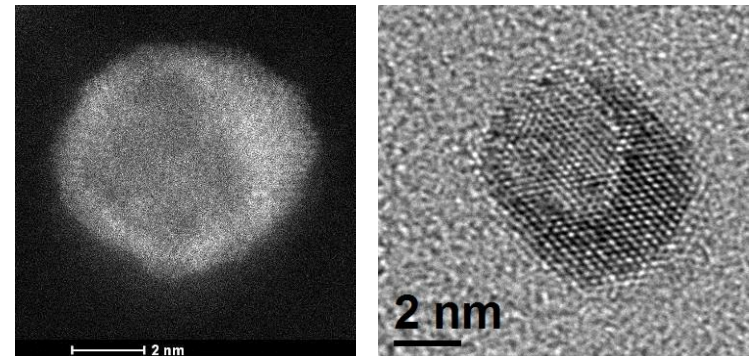
Ex. FeRh



A. Hillion *et al.*, *Phys. Rev. Lett.* **110**, 087207 (2013)

Ferromagnetic order stable at low T
(instead of anti-ferromagnetic)

Ex. CoAu

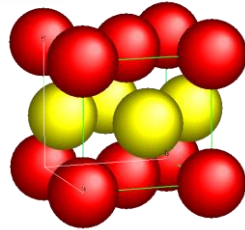


Original structures, magneto-plasmonic
interest

The case of CoPt and FePt nanoparticles

L₁₀ phase

- Chemically ordered
- tetragonal cell ($c/a < 1$)



The L₁₀ phase has a huge **magnetic anisotropy** constant ($K_{\text{eff}} \sim 5 \text{ MJ/m}^3$ for bulk CoPt)

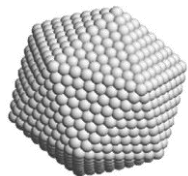
➔ Interesting for magnetic storage applications

The L₁₀ phase is stable at room temperature, but A1 is metastable

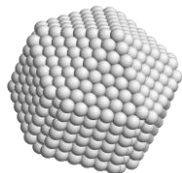
Open questions about the structure and magnetic properties of small nanoparticles:

Chemical order vs. size, finite size effects on the structure (defects, relaxation...), nanoalloy effects...

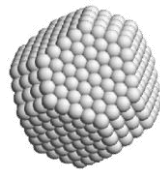
➔ It is a challenge to determine the intrinsic magnetic properties of chemically ordered particles



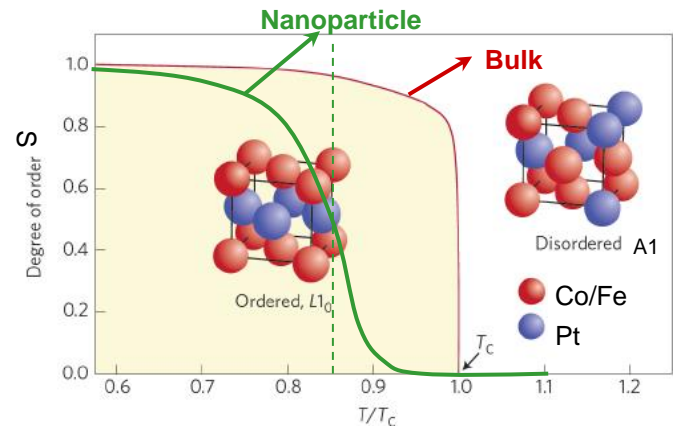
Icosahedron



Decahedron

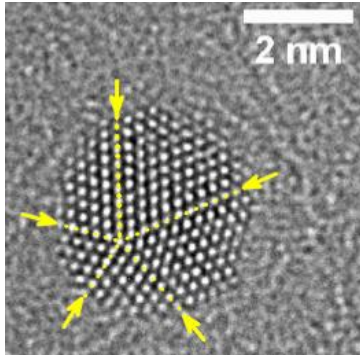


Truncated-octahedron

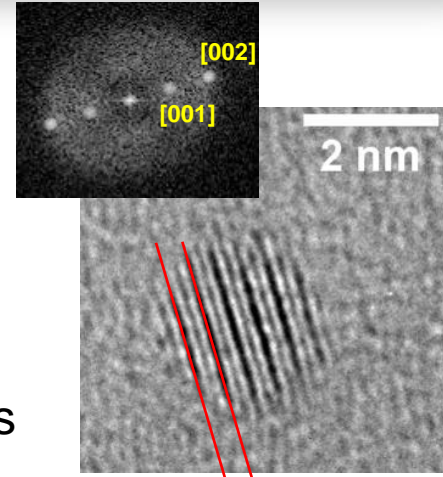


N. Blanc *et al.*, Phys. Rev. B **83**, 092403 (2011)

F. Tournus *et al.*, Phys. Rev. Lett. **110**, 055501 (2013)



- ✓ Coexistence of fcc and multiply-twinned particles
- ✓ No chemical order before annealing
- ✓ L1₀ contrast ([001] peak) after annealing, even for the smallest particles

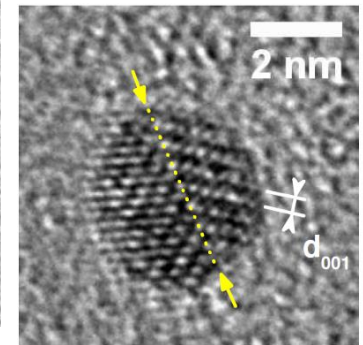
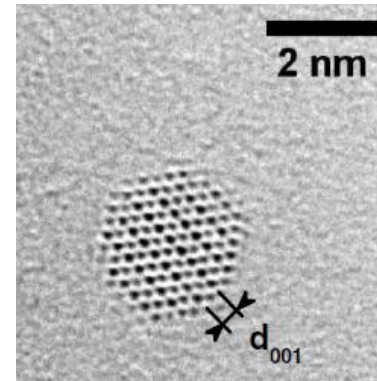


- ✓ Single L1₀ domain or several domains (twins, variants)

Decahedral particles with a chemical order

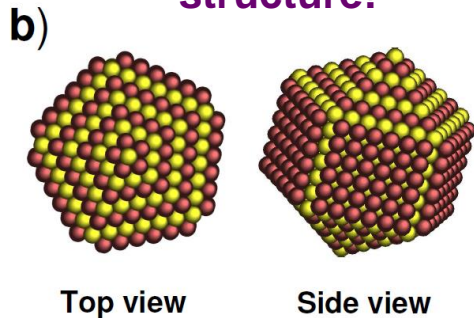
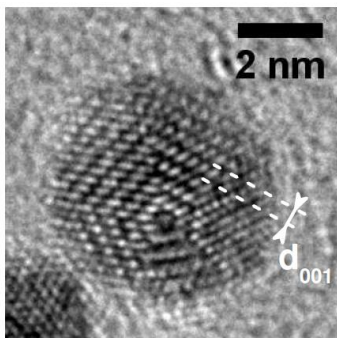
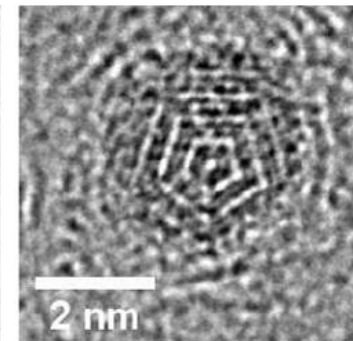
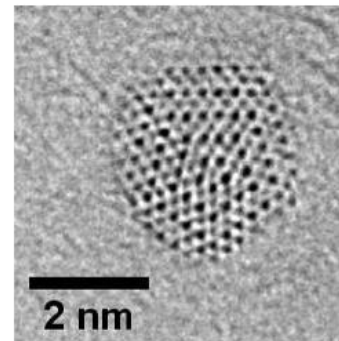
➔ Five L1₀ domains with c axes in different directions

Theoretically predicted structure!



$d_{001} \sim 3.7 \text{ \AA}$

FePt

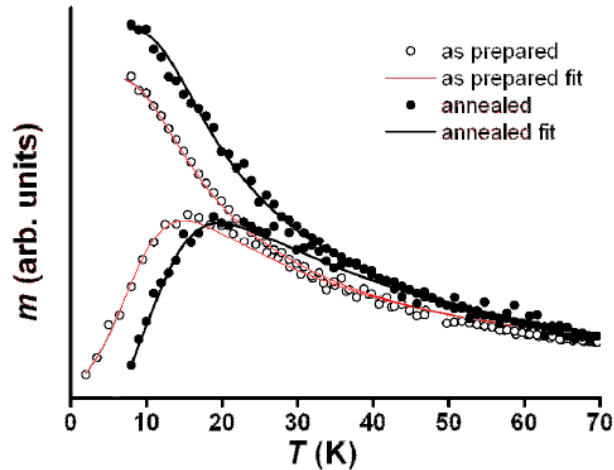


Evolution of the magnetic properties (SQUID, XMCD measurements) upon chemical ordering but...

The magnetic anisotropy of CoPt nanoparticles remains much smaller than the bulk (of the order of 300 kJ/m³ instead of 5 MJ/m³)

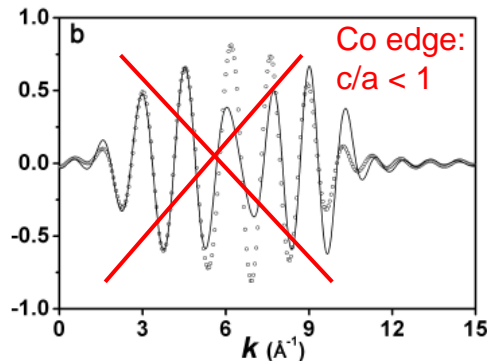
F. Tournus *et al.*, Phys. Rev. B **77**, 144411 (2008)

F. Tournus *et al.*, Phys. Rev. B **81**, 220405(R) (2010)



Possible explanations:

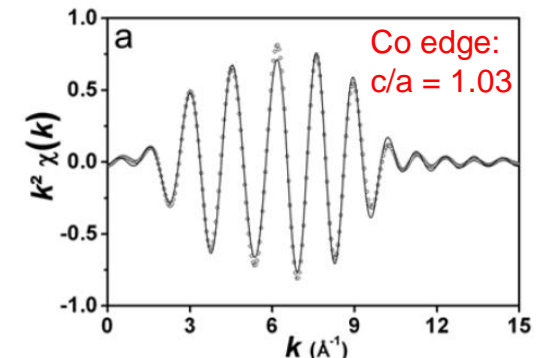
- ✓ Particles with several L1₀ domains
➡ **Lowering of the anisotropy!**
- ✓ Coexistence of various structures
➡ Anisotropy constant dispersion
- ✓ Relaxation of interatomic distances due to finite size: $d_{\text{Pt-Pt}} \neq d_{\text{Co-Co}}$



EXAFS measurements,
DFT calculations



Tetragonalization
different from the bulk



N. Blanc *et al.*, Phys. Rev. B **87**, 155412 (2013)

V. Dupuis *et al.*, Eur. Phys. J. B **86**, 1 (2013)

Quantitative analysis of experimental curves

➡ **Best fit procedure**

- ✓ Complementary measurements, analyzed with a combined fit (shared parameters)
 - Low field susceptibility (ZFC/FC)
 - Superparamagnetic magnetization loop
 - Low temperature hysteresis loop
 - Isothermal remanent magnetization (IRM)

- ✓ Efficient modelling for realistic assemblies of particles (**dispersion of size and anisotropy**, temperature...)

➡ Ingredients: Stoner-Wohlfarth (macrospin), Néel relaxation

Negligible inter-particle interactions?

➡ Intrinsic magnetic properties, signature of the individual nanomagnets

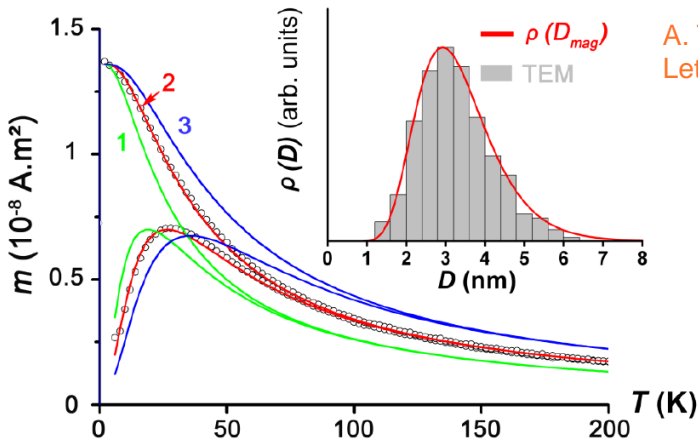
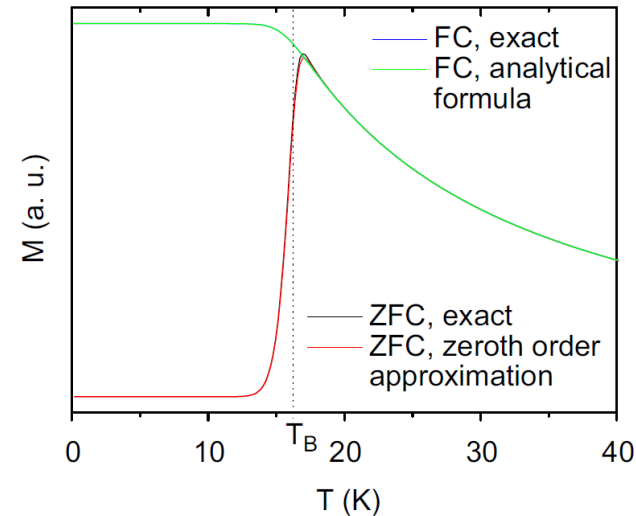
- Test of the theoretical models (macrospin, uniaxial anisotropy etc.)
- **Accurate determination of magnetic parameters (size and anisotropy...)**

Efficient simulation of the entire ZFC/FC curves for an assembly with a particle size distribution

- ✓ “Progressive crossover model”
- ✓ Semi-analytical calculations

ZFC/FC curves sensitive to the magnetic size distribution

➔ A combined fit is more discriminating



A. Tamion et al., Appl. Phys. Lett. **95**, 062503 (2009)

Beyond the uniaxial anisotropy approximation

➔ Bi-axial contribution
+ anisotropy constant dispersion

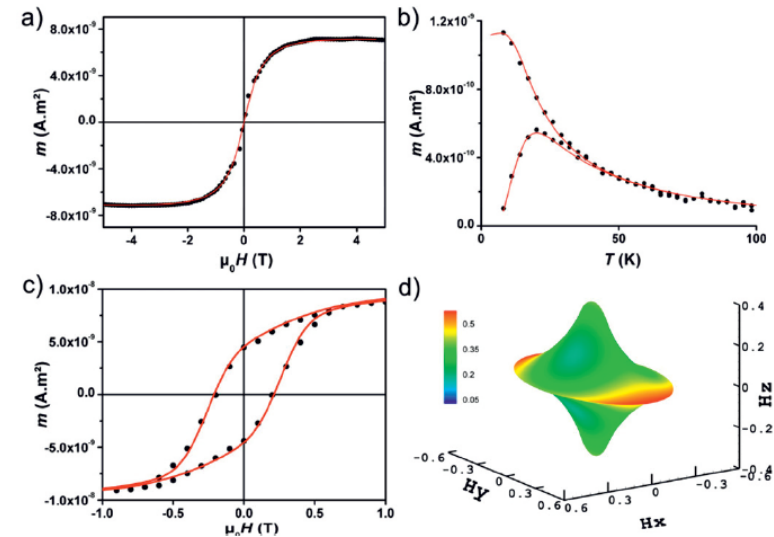
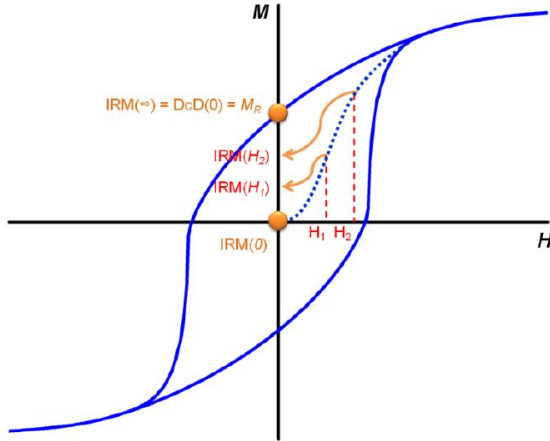


Fig. 4. (Color online) Hysteresis loops at 300 K (a), at 2 K (c) and ZFC/FC (b) for annealed CoPt nanoparticles embedded in C matrix. The solid lines correspond to the fit. Mean astroids associated to the biaxial fit (d).

Isothermal remanent magnetization



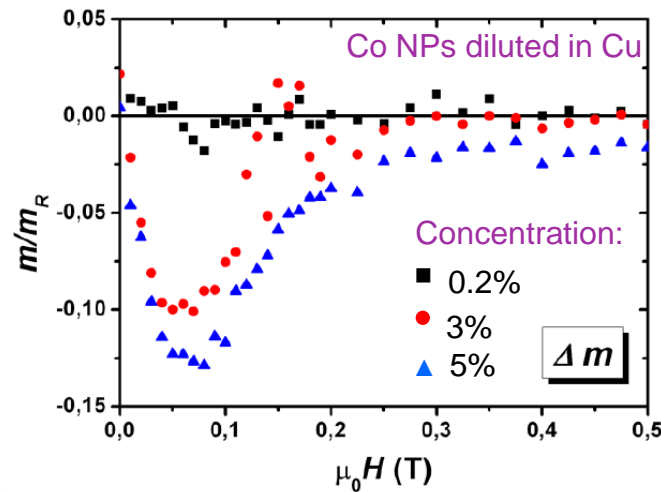
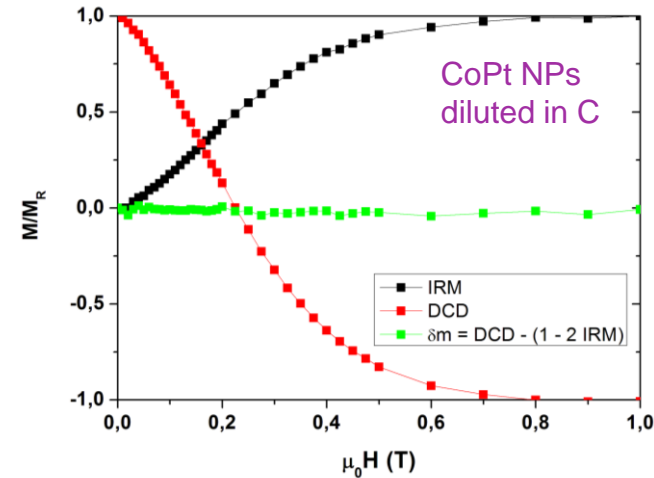
Signature of **irreversible** magnetization switching



Δm parameter:

$$\Delta m = DcD/m_R - (1 - 2 \text{IRM}/m_R)$$

If no interaction:
 $\Delta m = 0$ verified



Δm is very sensitive to interactions!

- ✓ Modelling of IRM curves, without interactions
- ➔ Fit of experimental measurements possible!

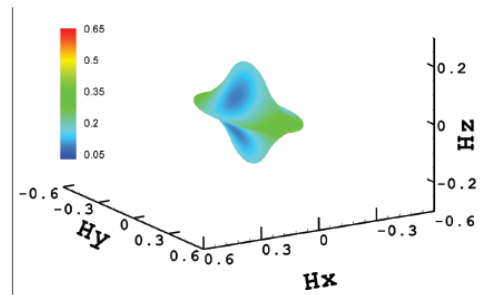
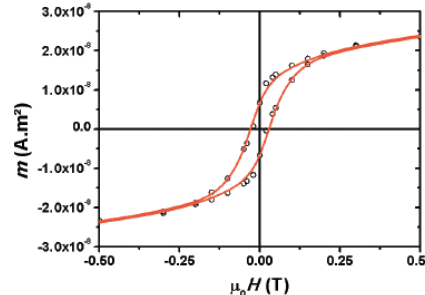
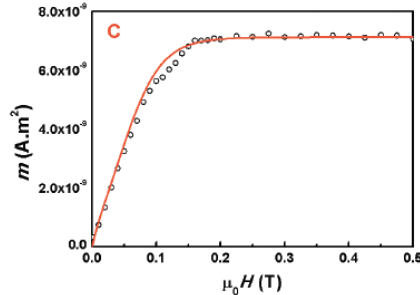
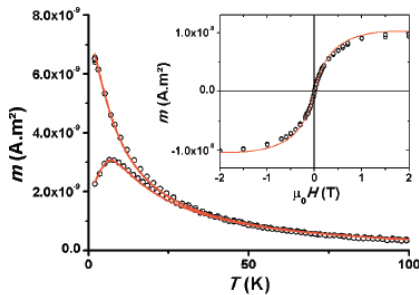
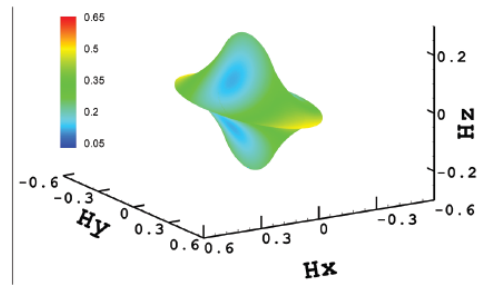
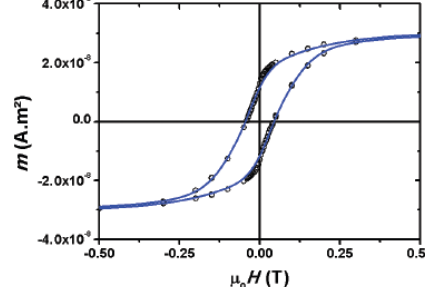
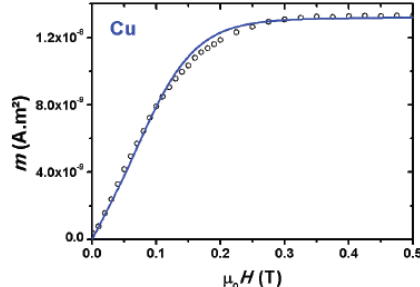
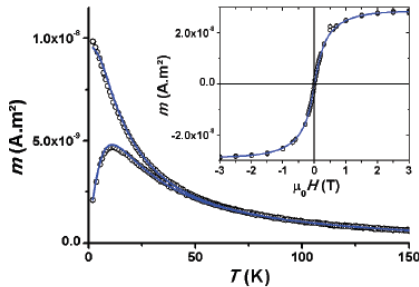
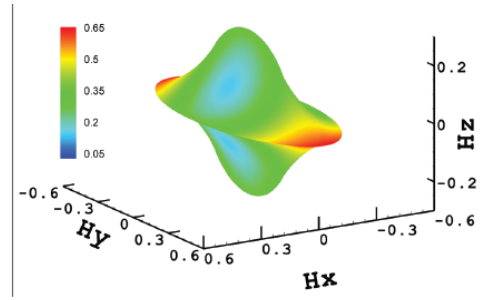
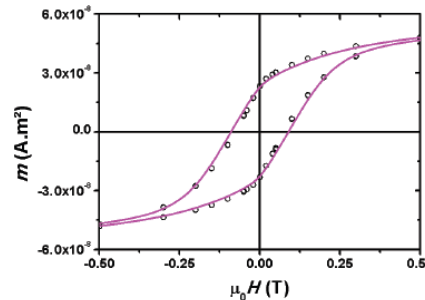
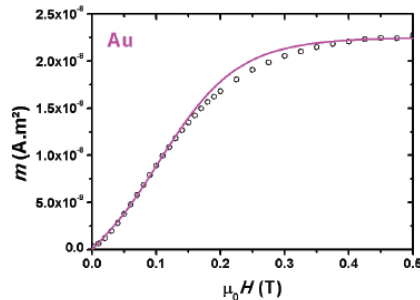
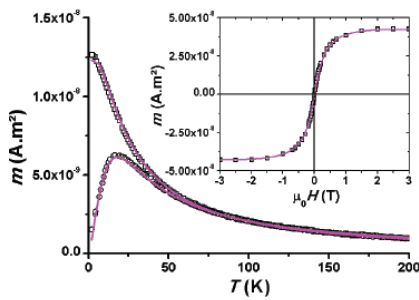
F. Tournus, J. Magn. Magn. Mater. **375**, 194 (2015).

Different physical processes ➔

IRM and ZFC/FC curves are complementary

Global fit

- Accurate determination of the “intrinsic” magnetic properties
- Original results with conventional magnetometry on nanoparticles



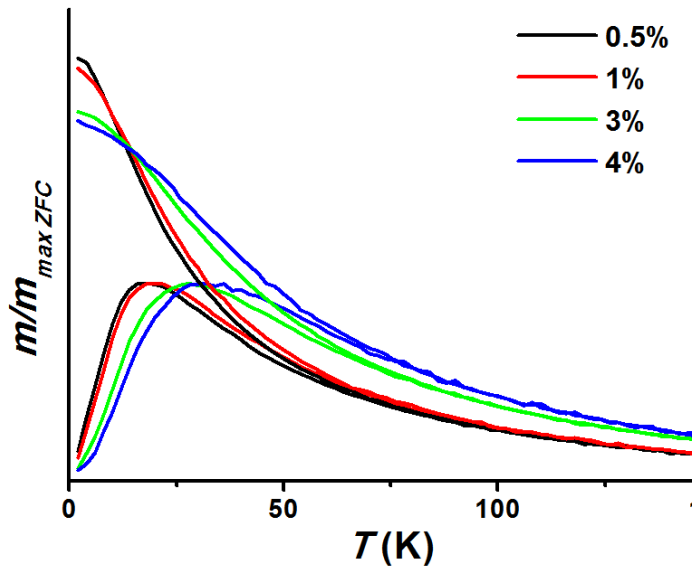
Co nanoparticles in gold, for increasing concentration (low interaction regime, only a few percent of volume concentration)

A. Hillion et al., Phys. Rev. B **95**, 134446 (2017)

- Modification of the IRM/DCD curves



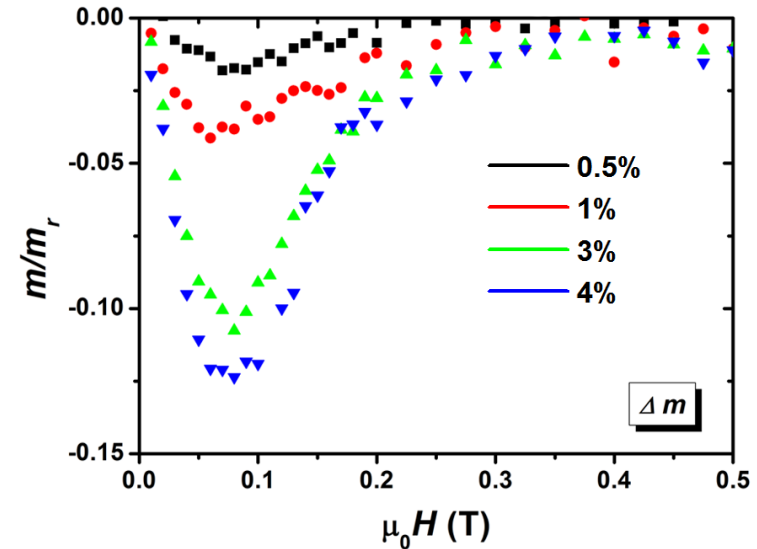
Significant peak (<0) for the Δm parameter



- Impact on ZFC/FC



Increase of the ZFC peak temperature!



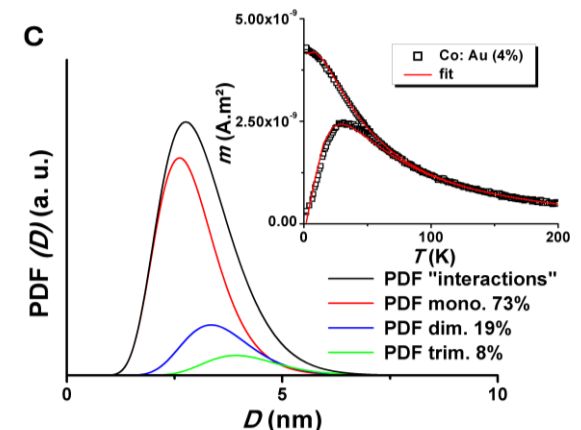
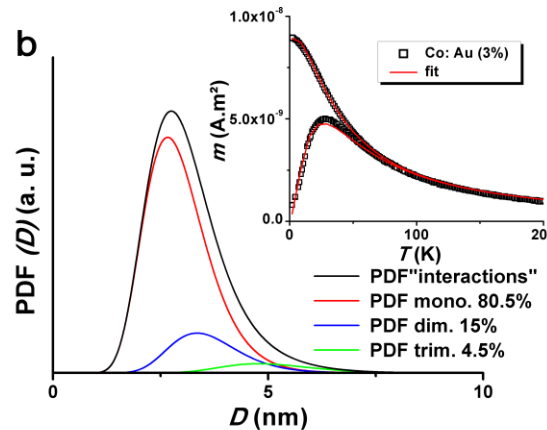
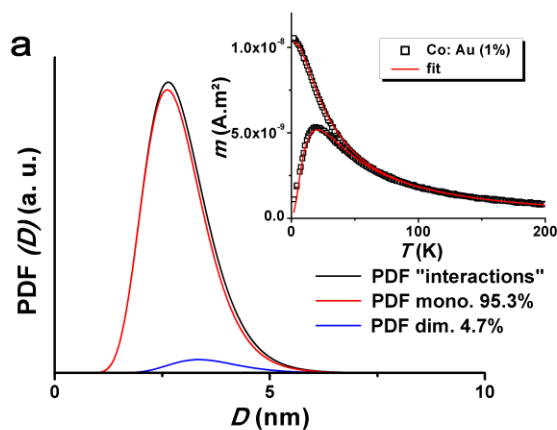
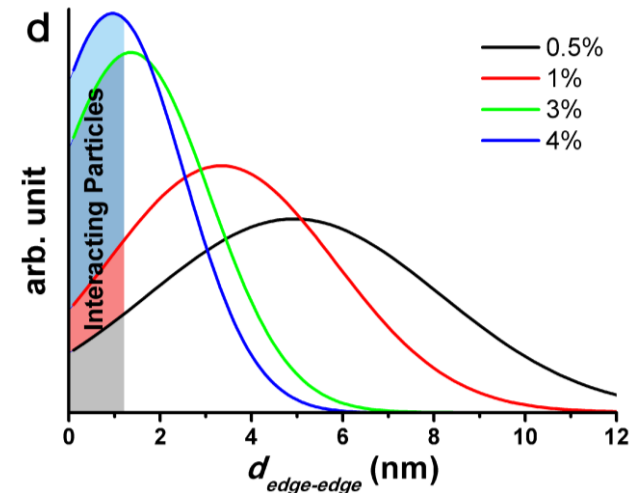
- ✓ We know the properties of the deposited particles
- ✓ We know the nearest-neighbor (NN) distance distribution
- ➡ Particles can interact if they are close enough from each other, which can modify the effective particle size distribution

✓ Under a given **interaction length ℓ^*** particles form dimers (or trimers)

➔ The proportions of **magnetic multimers** directly depend on the concentration

Sample concentration	T_{max} (K)	$\mu_0 H_c$ (mT)	m_r/m_s	x_{dim} (%)	x_{trim} (%)
0.5 %	17	85	0.38	2	0
1 %	19	75	0.38	4.7	0.03
3 %	28	75	0.38	15	4.5
4 %	32	70	0.31	19	8

Statistical distribution of NN distances



✓ Interaction length $\ell^* = 1.2 \text{ nm}$ for Co nanoparticles in Au: account for ZFC/FC evolution



“Super-ferromagnetic” dimerization
(Exchange-like, RKKY...)

✓ In Cu, similar to Au

➔ Shift of the ZFC peak

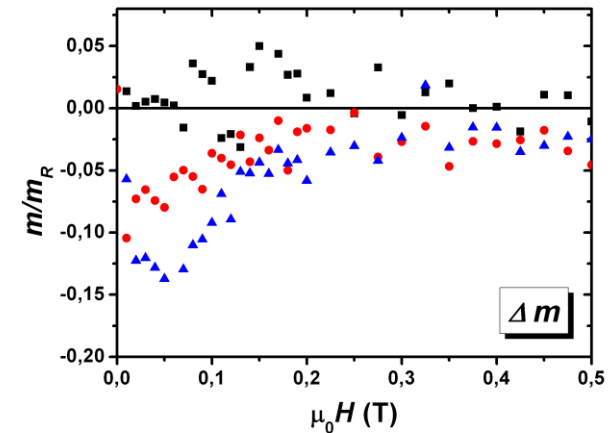
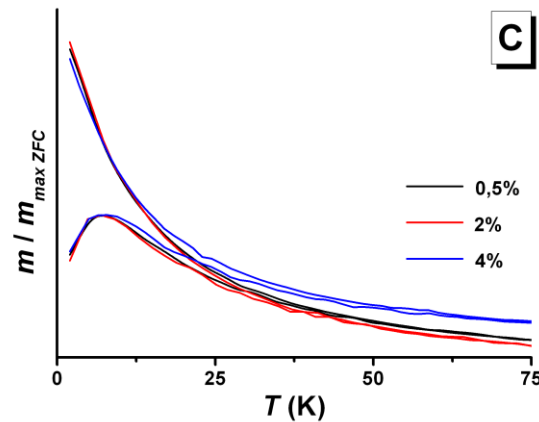
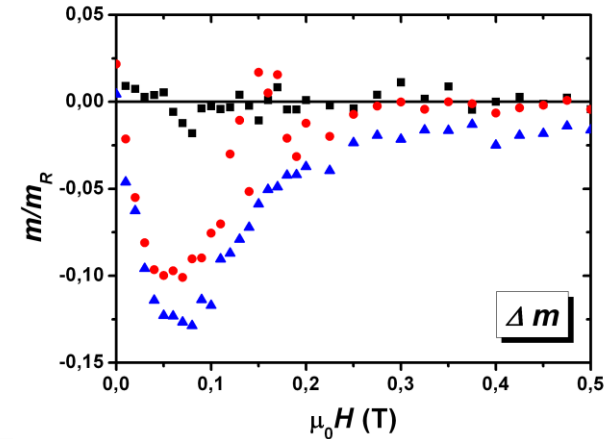
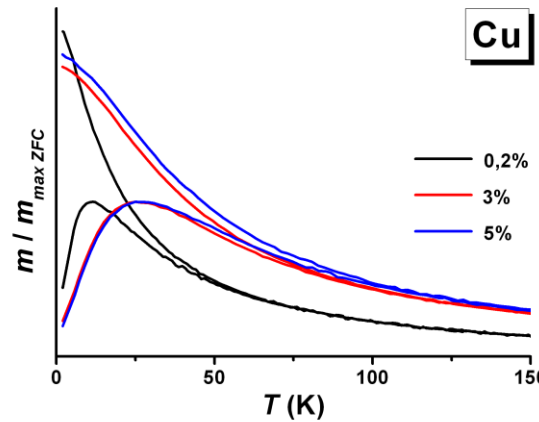
✓ In C, almost no change of the ZFC/FC

➔ But similar Δm evolution!

Very different behavior in metallic matrices and C

{

Co NP @ Cu: $l^* = 1.3$ nm
Co NP @ C: $l^* = 0.3$ nm



- ❑ ZFC/FC sensitive to slight magnetic size changes
- ❑ Δm highly sensitive to dipolar interactions

For higher concentration, the model is not valid anymore ➔ collective effects...

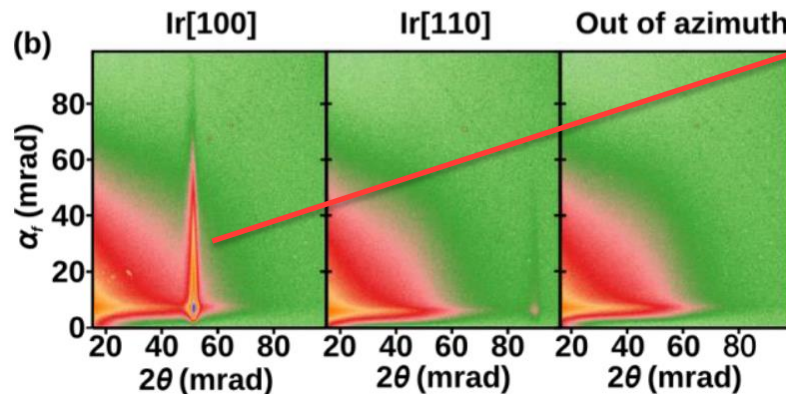
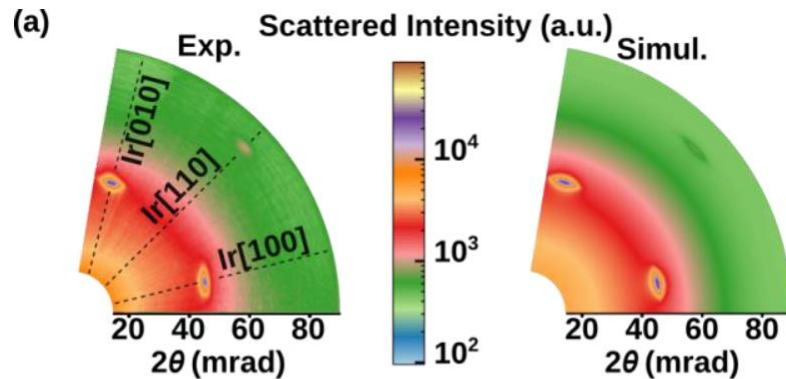
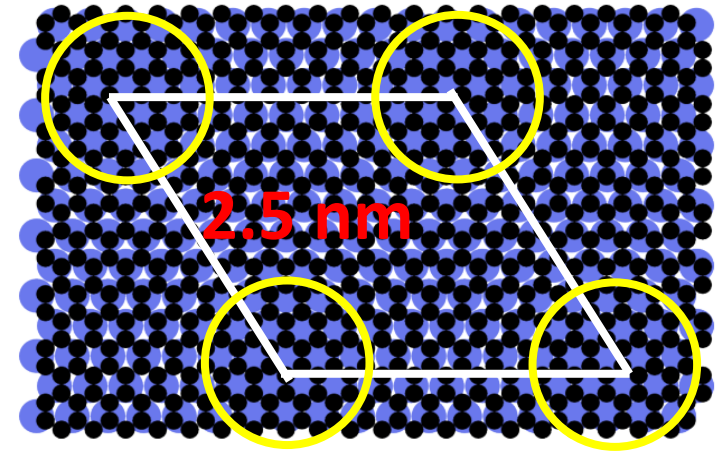
Outline

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- Specific behavior of bimetallic nanoparticles
- Analysis of magnetic measurements: intrinsic properties with diluted nanomagnet assemblies and beyond...
- Interface effects with nanomagnets on cristaline surfaces: FePt/graphene/Ir(111) and FeRh/BaTiO₃
- Nano-composites: Co clusters in a FePt matrix

- Moiré pattern with graphene on Ir(111) (hexagonal lattice, $a=2.5$ nm)

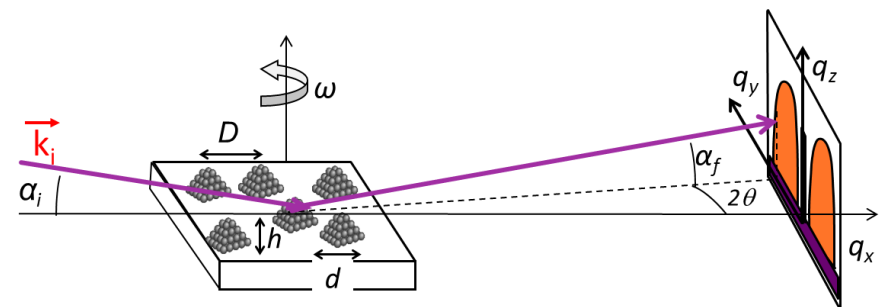
➔ Use as a **template** for NP organization: preferential pinning sites

Graphene on Ir(111)



- Successful self-organization of small Pt nanoparticles deposited by LECBD
- ➔ Organization probed by GISAXS

Correlation peak (in Moiré directions), signature of coherent organization



Grazing Incidence Small Angle X-ray Scattering

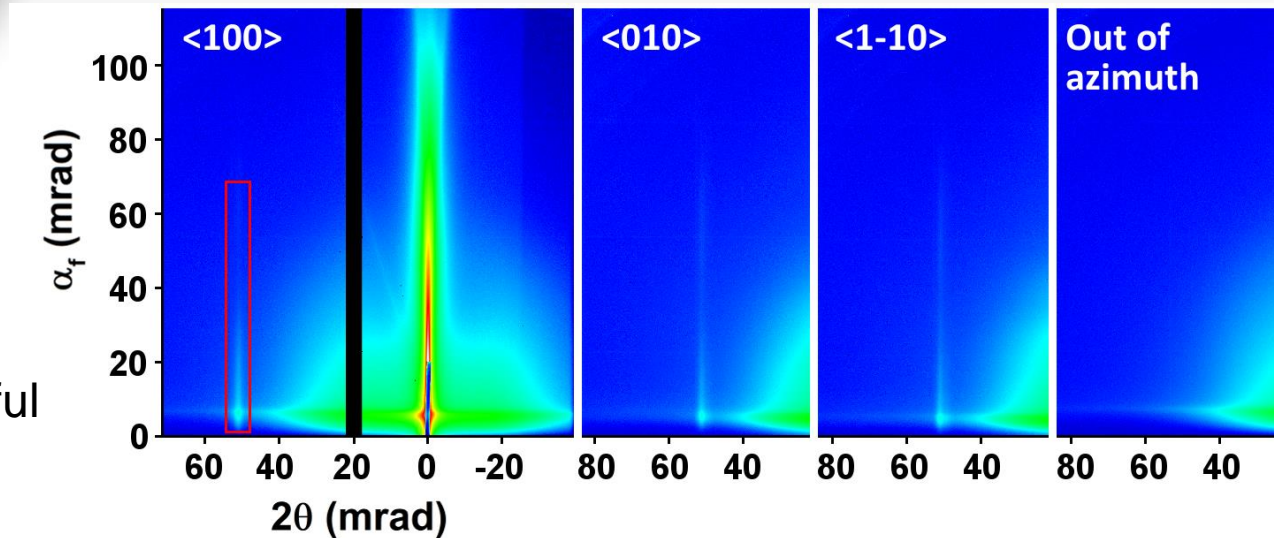
FePt, 2.2 nm

a-C

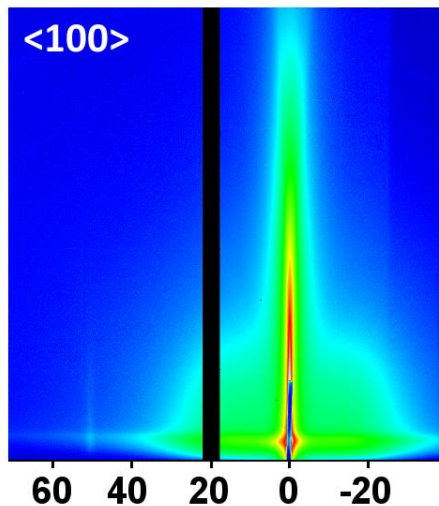


- ✓ Same approach successful for size-selected **FePt** nanoparticles

➡ (Partial) organization on the Moiré sites



Rk.: Organization was not possible using atomic deposition



- ✓ A signature of spatial organization is preserved up to 600°C
- ➡ Annealing compatible with L1₀ chemical ordering

Quantitative analysis is still in progress...

- ✓ Sample also characterized by XMCD
- ➡ Magnetic properties

XMCD (Fe L edges)

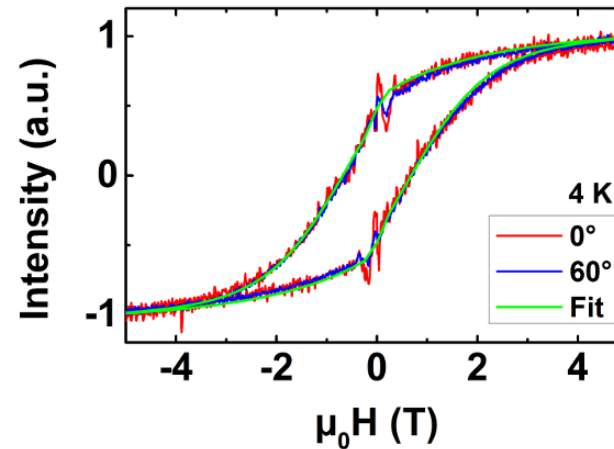
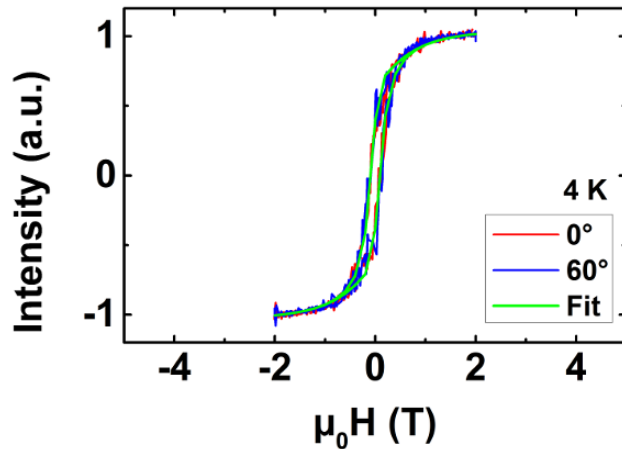
FePt, 2.2 nm

a-C



Before annealing

After annealing



✓ No effect of incidence angle: isotropic behavior

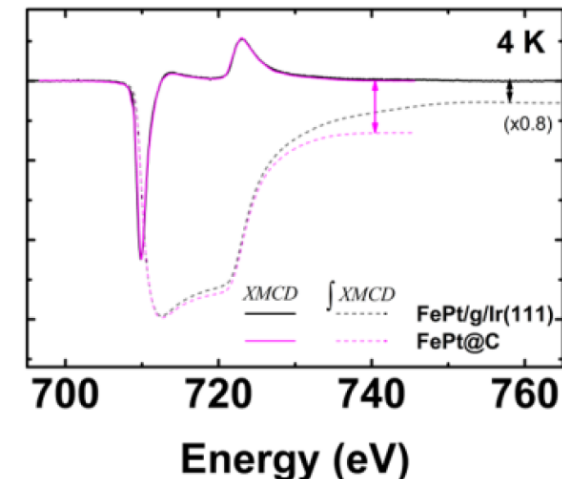
✓ Huge increase of coercivity: particles with a magnetic anisotropy larger than 1 MJ/m^3 (and large dispersion)

➡ Must reflect $L1_0$ chemical ordering of FePt NPs

✓ In addition, evolution of Fe atomic moments

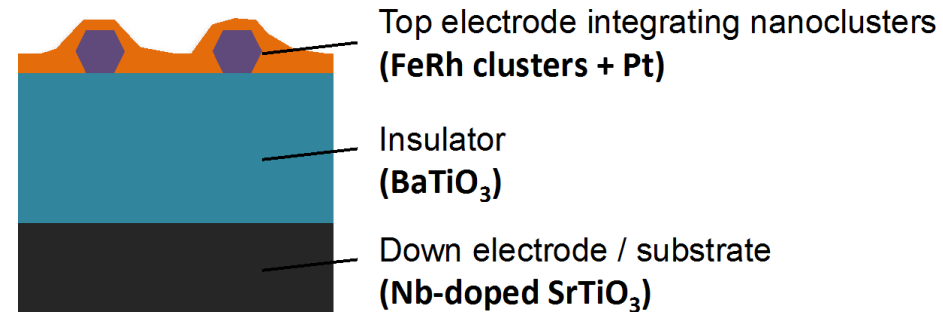
- Large value of m_S before annealing (as for the bulk)
- Lower m_L value than for particles in carbon matrix

➡ Interface effect?



Starting project

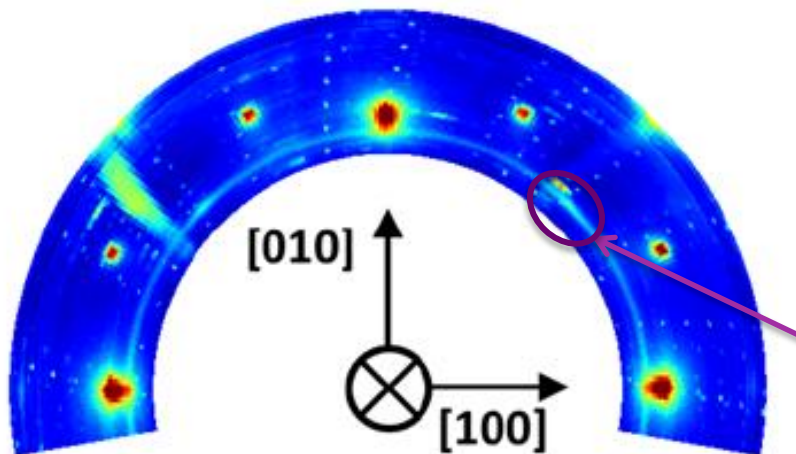
FeRh nanoparticles deposited on crystalline BaTiO₃ surface (ferro + piezoelectricity)



Electric field control of NP magnetic properties

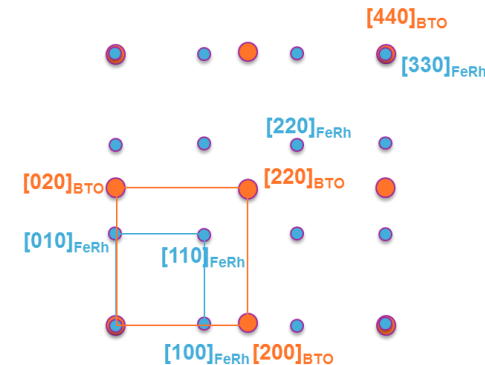
→ FeRh sensitive to charge effects and distortions

- ✓ First results on the structure (GIXRD):
 - Chemically ordered B2 phase after annealing
 - Epitaxial relationships



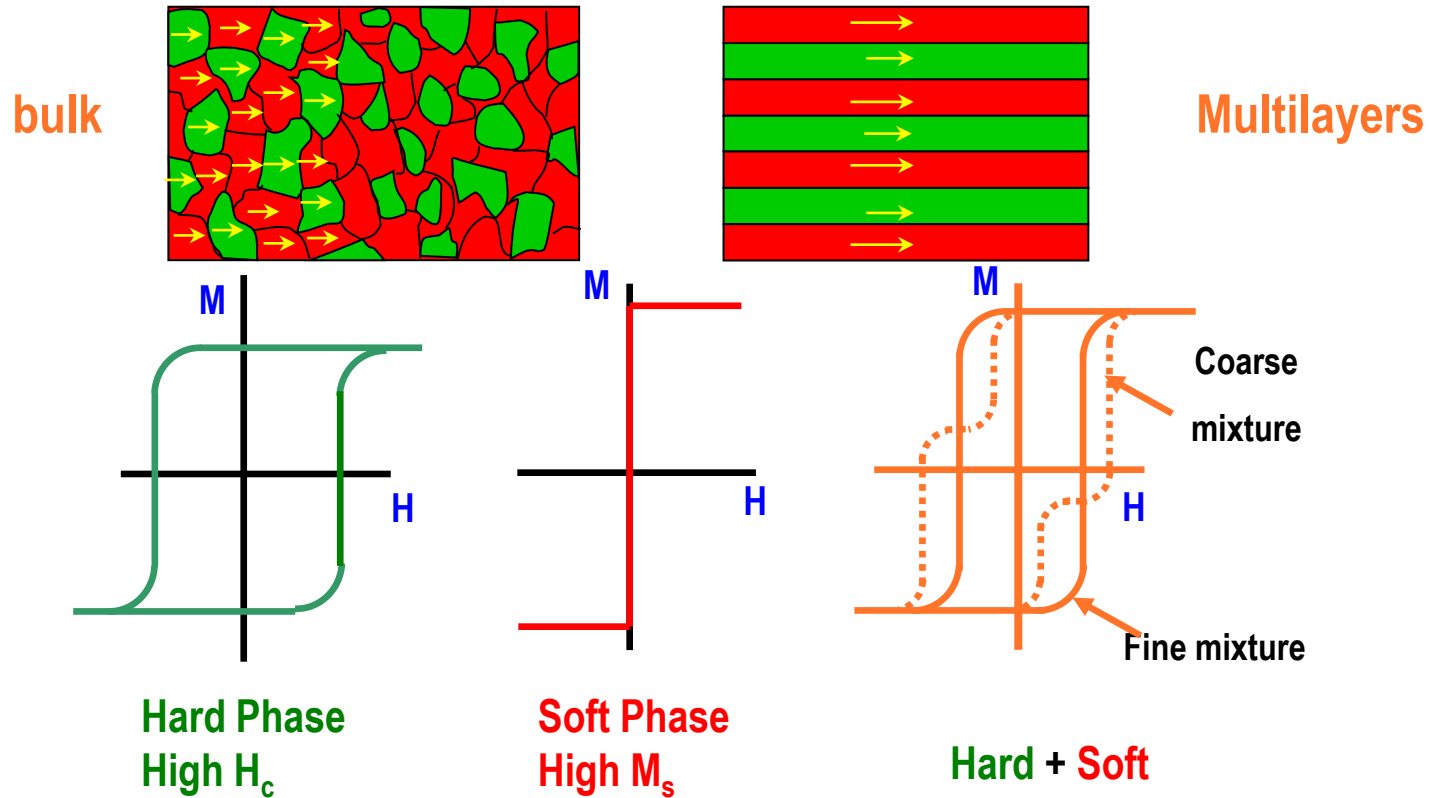
Intense signal of [110]_{FeRh}

Exotic epitaxy (not observed for thin films) in addition to the usual one!



Commensurability: $2 [220]_{\text{BTO}} = 3 [110]_{\text{FeRh}}$

- **Principle:** Fine mixture of soft (high M_S) and hard (high K_u) phases for enhanced remanence and energy product
- **Requirement:** Nanosized and isolated soft grains (typical sizes < 10 nm)

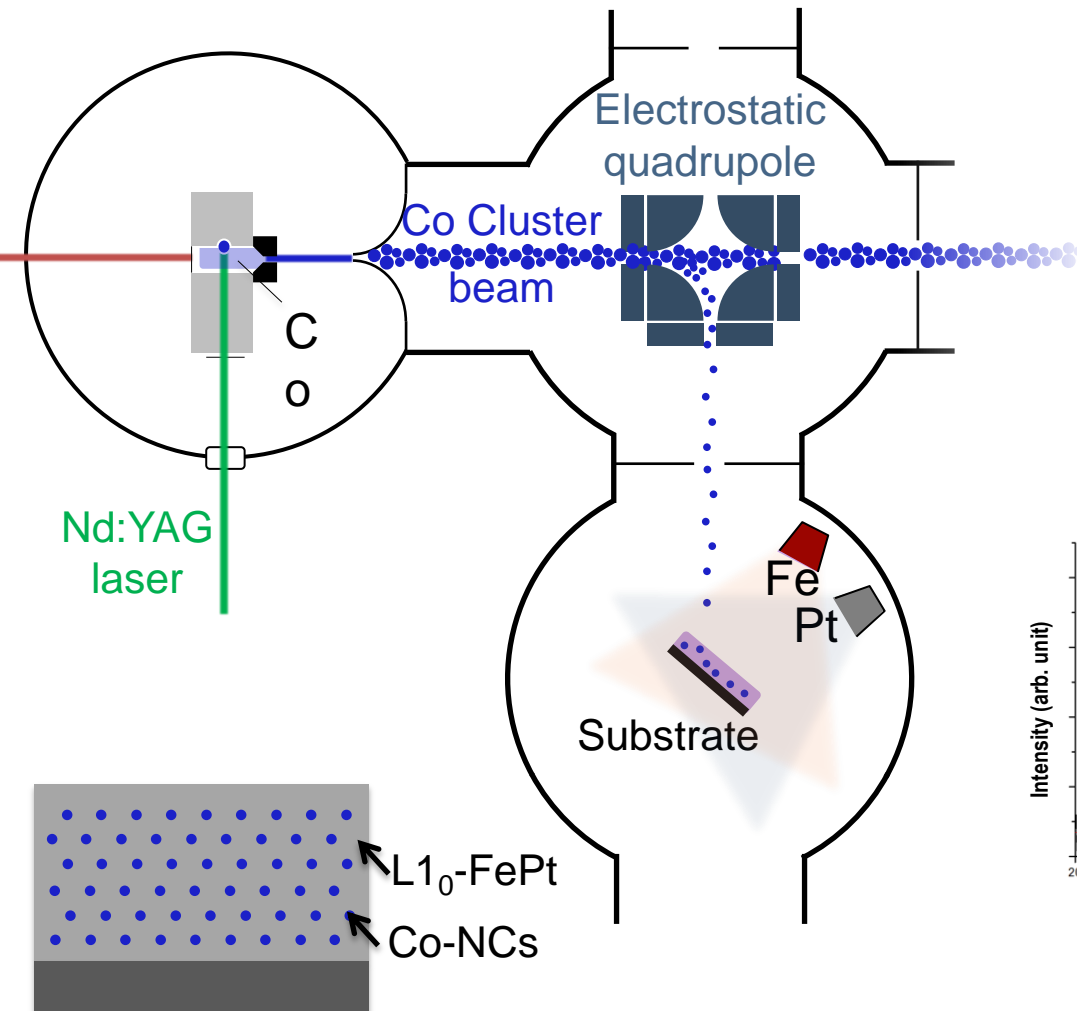


- **Our approach (by cluster deposition):**

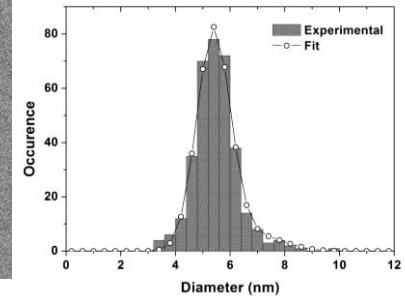
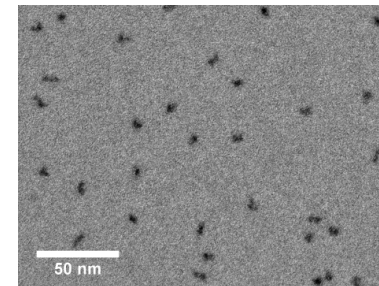
Nanocomposite films made of $L1_0$ -FePt with embedded Co nanoclusters

ANR « SHAMAN », Partners: I. NEEL (Grenoble), ESRF, ID12 (Grenoble), SPCTS (Limoges)

- **LECBD + e beam evaporators:** Sequential deposition of Co-NCs, Fe and Pt

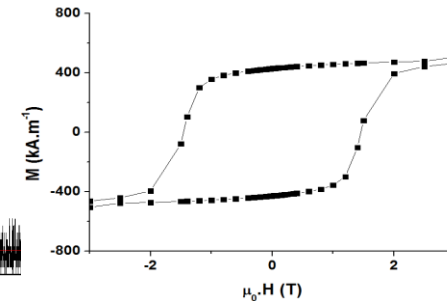
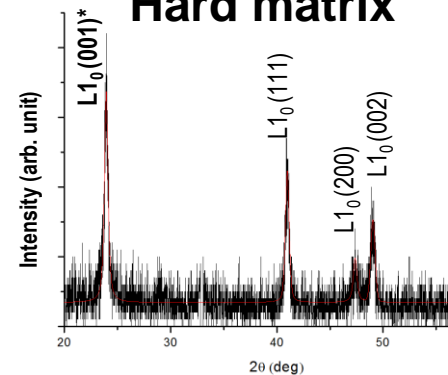


Soft nanosized grains



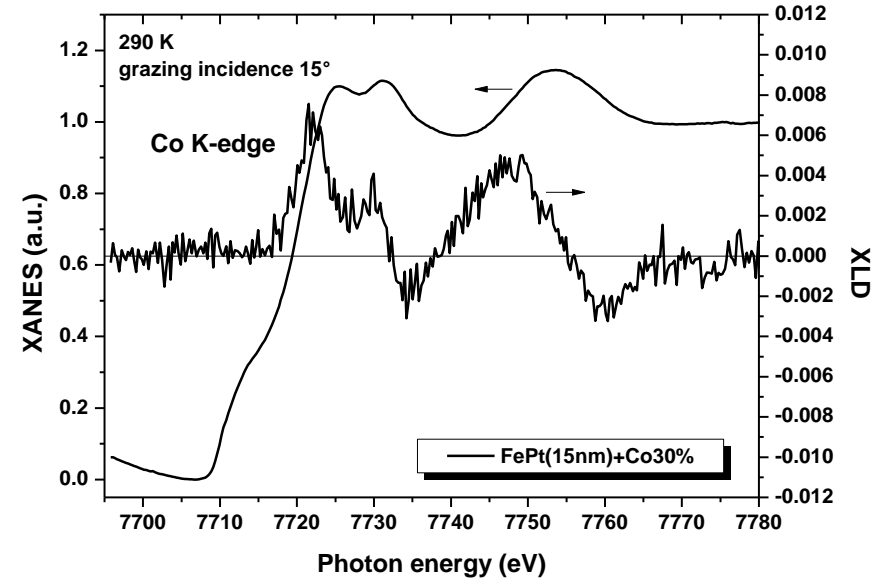
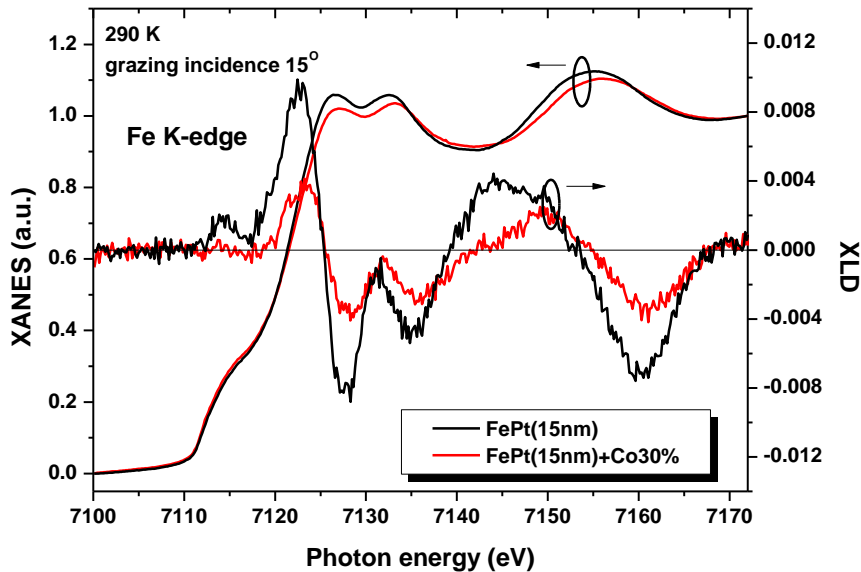
6nm Co nanoclusters (fcc)

Hard matrix



L1₀-FePt films obtained from {Fe/Pt} multilayer growth

✓ **First results with XAS at Fe and Co K-edges**



Effect of Co-NCs:

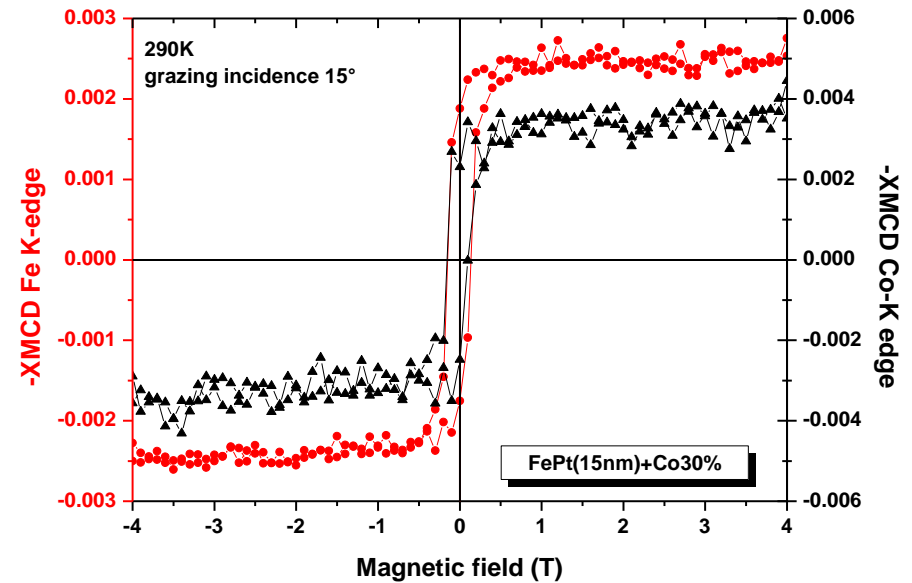
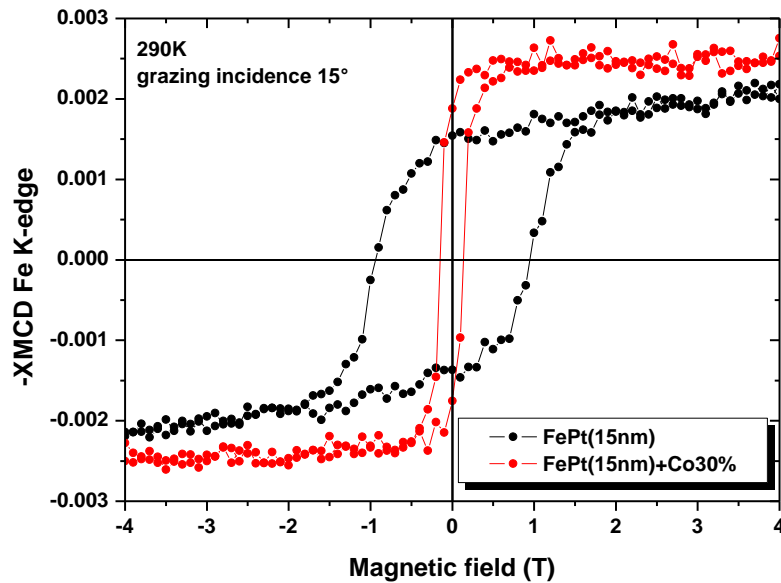
- (XANES) reduced lattice parameter on Fe environment
- (XLD) lower degree of texture

Tetragonal environment for Co atoms:
from fcc to fct



Co atoms enter in the textured FePt phase

✓ First magnetic results with XMCD (K-edges)



Effect of Co-NCs:
 H_C decreases ($\mu_0 H_C$ from 1.0 T to 0.2 T)
 m_{Fe} increases

Same reversal for Co and Fe moments

➡ Co and Fe moments are coupled
 Single phase behaviour

Our motivation: How can the magnetic properties of nanostructures be controlled and tailored?

➡ Exotic effects with magnetic nanoparticles

Our approach: Well-defined and original samples by cluster deposition

➡ UHV deposition, size-selection, diluted assemblies, clusters on surfaces...

Our expertise: Investigation of small nanomagnets, from individual objects to complex systems!

➡ Magnetometry measurements and modelling, structural characterizations, synchrotron experiments...

Thank you!